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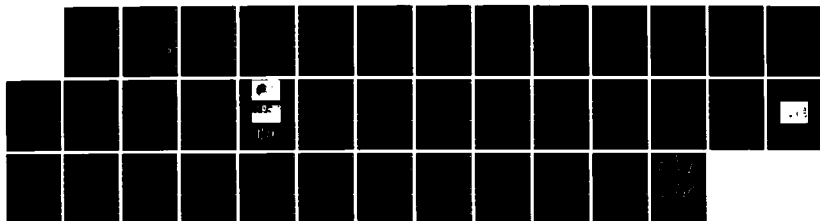
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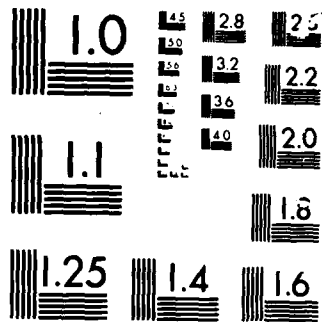
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INVESTIGATION OF SINGLE CRYSTAL
SILICATES FOR BLUE TUNABLE LASERS

by

R. F. Belt and J. A. Catalano

Quarterly Report No. 1
September 1, 1985 - November 30, 1985
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Airtron Division
Litton Industries, Inc.
200 E. Hanover Avenue
Morris Plains, N. J. 07950

January, 1986

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Cathodoluminescence experiments on both powders and single crystals showed wavelength shifts with different rare earth ions. The Gd, Y, and mixed crystals of Gd-Y silicates gave good bright emission with Ce^{3+} . The emission is broad band and covers the wavelength range of 370-525nm. The peak emission can be centered near 450-460 by other compositions. Larger single crystals for laser hosts are under development.

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FOREWORD

This quarterly report describes research efforts on the preparation of single crystal rare earth and other silicate laser hosts. All crystals were doped with Ce^{3+} in an attempt to obtain blue or blue-green lasers which may operate near 450-480nm. The report relates work started in the period of September 1, 1985 up to November 30, 1985. All work was done under a program of Dr. Van O. Nicolai of the Office of Naval Research. It is now continuing under the administrative supervision of Dr. Robert Junker of the Office of Naval Research and technical monitoring of Dr. Leon Esterowitz at the Naval Research Laboratory.

All physical preparations were completed in the laboratories of Airtron Division of Litton Industries, 200 E. Hanover Avenue, Morris Plains, New Jersey, 07950. The program was directed by Dr. Roger F. Belt and Mr. Joseph Catalano served as Senior Staff Scientist. Mr. Gregory Foundos was the technician. Optical tests were performed at the Naval Research Laboratory through the courtesy of Dr. Leon Esterowitz. The X-ray patterns were run by Mr. John Yorston. Cathodoluminescent tests were performed by Dr. John Ings. The report was prepared by Dr. Roger Belt and Mr. Joseph Catalano. It was released for publication in February 1986.

1.0 Introduction

It has been a long term goal of the Navy to obtain high efficiency blue or blue-green solid state lasers. One of many approaches for tunable systems involves the luminescent 5d - 4f transitions of Ce^{3+} . A major problem has been to find suitable single crystal hosts which have the following features.

1. Chemical and physical properties similar to other good hosts, e.g. Al_2O_3 , $Y_3Al_5O_{12}$, $YLiF_4$.
2. Ability to incorporate Ce^{3+} , Pr^{3+} , or other rare earth ions substitutionally.
3. A high band gap of about 6 ev to prevent excited state absorption of Ce^{3+} .
4. High radiative lifetime and fluorescent yield of Ce^{3+} .
5. The presence of Ce^{3+} in a suitable site and symmetry to luminesce at 450-500nm.

Most of the previous work under this program ⁽¹⁾ has been concerned with the perovskite structured host materials such as $LaAlO_3$. This compound is a distorted cubic crystal which leads to a hexagonal or orthorhombic structure. Efforts to obtain the stable cubic form of the crystal have been largely negative. The crystal of pure $LaAlO_3$ gives fluorescence with Ce^{3+} but the presence of twins and a phase transformation prevents good laser crystals from being produced. So far no single high quality crystals of cubic perovskites have been grown. Most of the difficulties are associated with phase

behavior, melting points greater than 2000°C, and presence of inclusions or twins. During the course of our work, a few runs of polycrystalline Gd_2SiO_5 doped with Ce^{3+} were produced for trial fluorescent experiments. This material showed some promise for ultraviolet excited emission and gave a broad band which peaked around 430-470nm. Since Gd_2SiO_5 and other rare earth silicates show some promise for shifting the emission with Ce^{3+} doping and can be grown as single crystals, this appears to be a fortunate start for a blue-green laser. The thrust of our research effort was concerned with silicates of the type R_2SiO_5 where R= any rare earth.

The rare earth silicates are interesting laser hosts which exhibit nearly all of the features of doping, mechanical properties, low cost, congruently melting behavior, and apparently a completely isomorphous structure from La to Lu. A description of various compounds and structures in the $\text{R}_2\text{O}_3\text{-SiO}_2$ systems is given in a general reference.⁽²⁾ The main compounds are 1:1 R_2SiO_5 , 2:3 $\text{R}_4\text{Si}_3\text{O}_{12}$ and 1:2 $\text{R}_2\text{Si}_2\text{O}_7$. A typical phase diagram⁽³⁾ is given in Figure 1; this figure is for $\text{Gd}_2\text{O}_3\text{-SiO}_2$ but all other rare earths are nearly identical. It is a general feature that only the 1:1 compounds of type R_2SiO_5 are all congruently melting with melting points in the range of 1900-2000°C. The remaining 1:2 or 2:3 type silicates are incongruently melting and special procedures must be used to obtain single crystals. Thus it appears that only the 1:1 composition may be grown large enough to yield traditional laser rods of 2-10cm length.

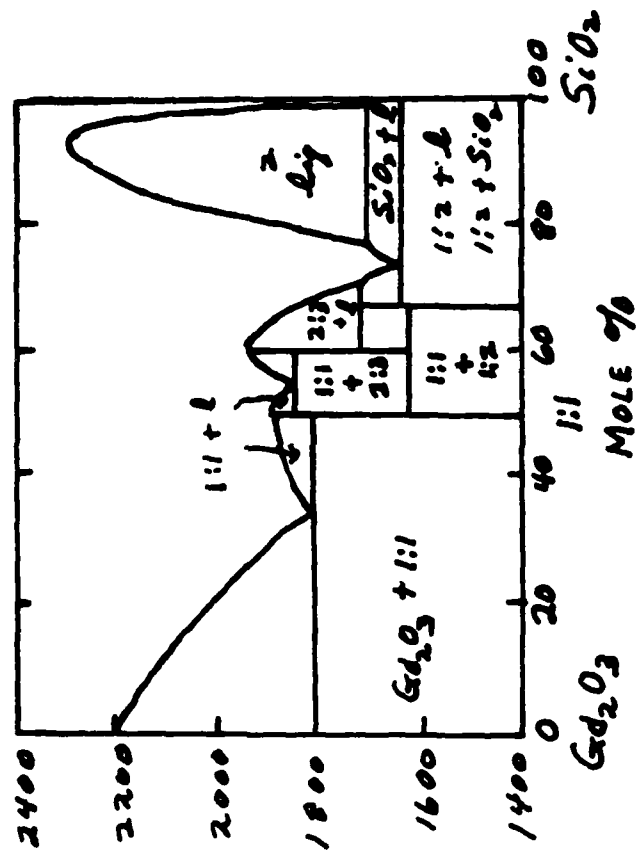


Figure 1 System Gd_2O_3 - SiO_2 . Oxide ratios of compounds are given as $Gd_2O_3:SiO_2$. (Ref. 3)

The few preliminary studies of rare earth silicate growth were all performed by the flux method. ⁽⁴⁻⁸⁾ This method utilizes fluxes such as Bi_2O_3 , $\text{Li}_2\text{O}-\text{MoO}_3$, KF , $\text{PbO}-\text{PbF}_3$, or combinations of the preceding to grow a particular composition of silicate. The crystals grown from these fluxes were of mm size and useful for X-ray or other structural studies. In all cases the absorption or emission spectra were not recorded. Neither were there any laser studies performed because of the size of the crystals. No further efforts were described to obtain single crystals for any purpose.

The earliest growth of Y_2SiO_5 single crystals was done in conjunction with laser research in the Soviet Union. In 1973 it was reported ^(9,10) that $\text{Nd}^{3+}:\text{Y}_2\text{SiO}_5$ was lased successfully at 300°K . Rods of mm size were used and the material had a low threshold of a few joules in a pulsed mode of operation. The emission spectrum, threshold, and other laser parameters were highly dependent on the orientation of the crystal axes of the laser rods. It appears that the crystal structure is orthorhombic and the two site symmetries at the Y^{3+} may be slightly different. What governs the site preference upon doping for laser action has not been explained. In Gd_2SiO_5 doped with Ce^{3+} , it may give rise to the small auxiliary peak near 545nm. In any case, a more detailed spectral study is required for the growth and doping of the single crystals.

In 1983, workers in Japan reported ⁽¹¹⁾ the Czochralski growth of $\text{Ce}^{3+}:\text{Gd}_2\text{SiO}_5$ for a scintillator application. These crystals were 25mm diameter and 40mm long. The luminescence spectrum of the 1.0 atomic percent Ce crystal is shown in Figure 2. The excitation is by 254nm ultraviolet light. It can be seen that the broad band emission has a peak in the blue near 450nm. The substitution of Y for Gd shifts the peak further into the blue near 400nm. While no experiment was performed on La addition, the larger unit cell and reduced crystal field about Ce^{3+} will probably shift the peak in the opposite direction, i.e. toward 470-480nm. The fluorescent lifetime of the Ce^{3+} in Gd_2SiO_5 is about 60ns. This value is equal to or slightly larger than Ce^{3+} in YAG or other oxide hosts.

Further literature on the Czochralski type growth is not available. In November, 1984, a personal visit was made to Philips Laboratories in Eindhoven, The Netherlands. It was learned that this laboratory was growing single crystals of Y_2SiO_5 by pulling from the melt. ⁽¹²⁾ The crystals were about 2 x 8cm and desired for phosphor research. In some previous papers by Philips personnel, the silicates were found to be some of the most efficient phosphors available. The work at Philips also confirms that good single crystals can be grown.

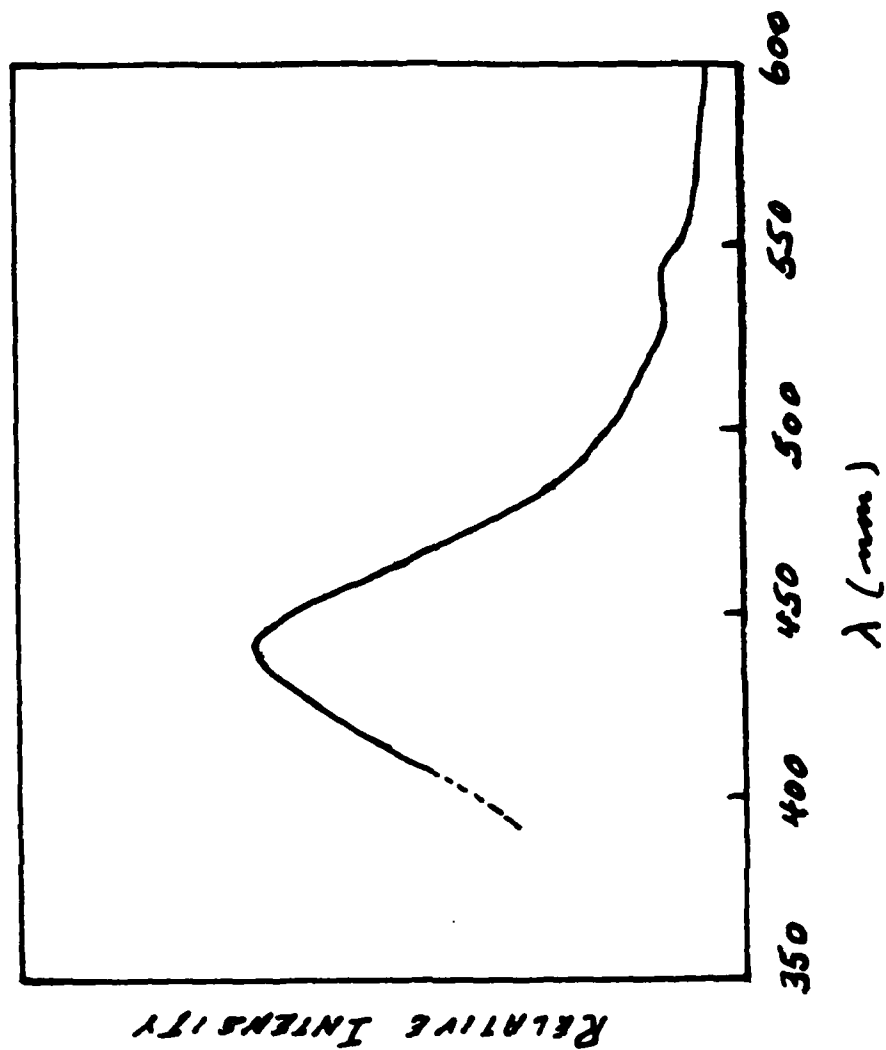


Figure 2 Luminescence spectrum of Ce:Gd₂SiO₅ single crystal excited by ultraviolet (wavelength 254nm) at room temperature. (Ref. 11)

2.0 Experimental Procedures

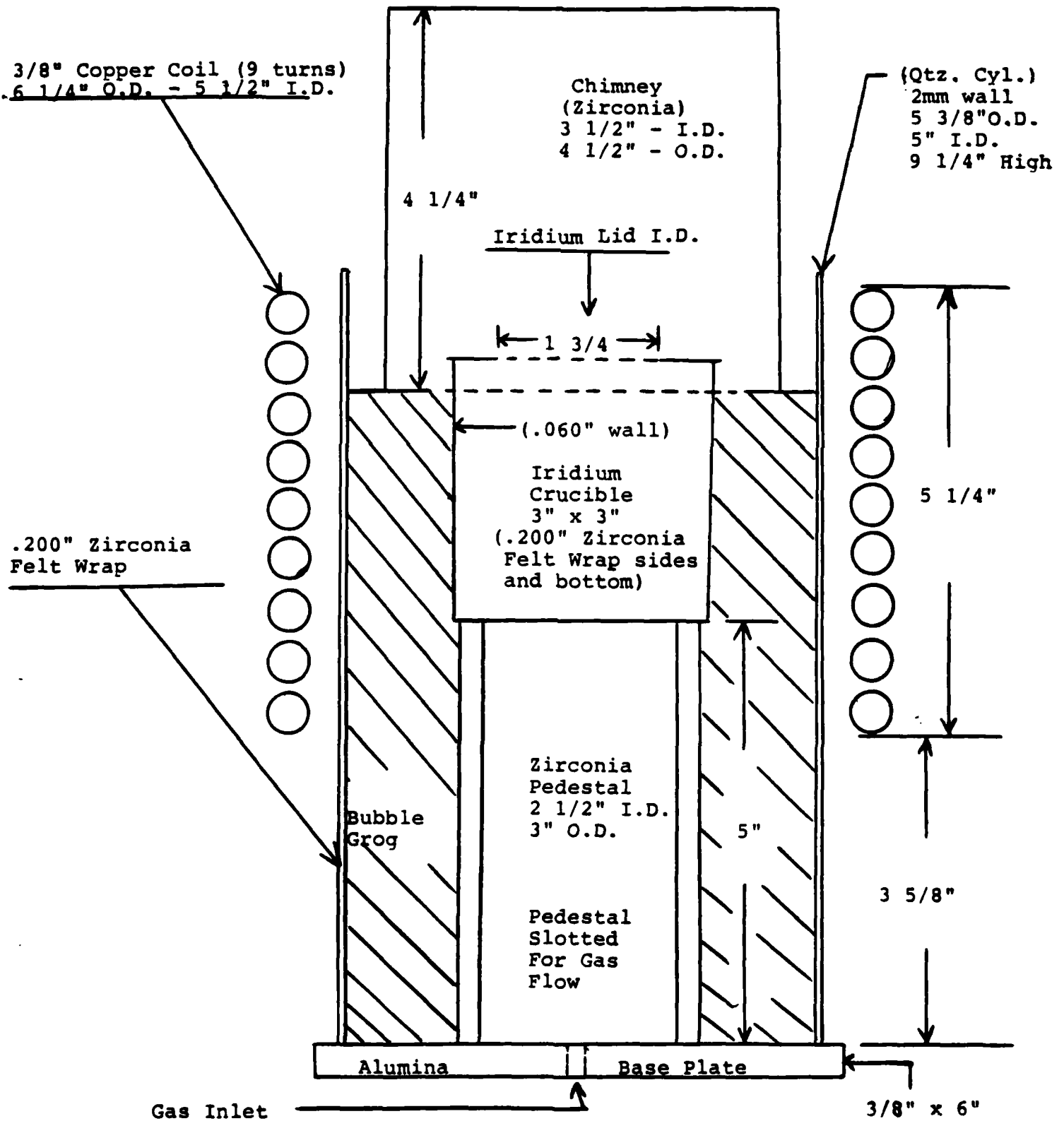
2.1 Feed Material

The precursor powders used in these experiments were of 4-9's purity or better. The CeO_2 was purchased from Spex Industries, Inc., the Gd_2O_3 and SiO_2 from the United Mineral Corporation, the Y_2O_3 from Rhone-Poulenc, and the La_2O_3 and Lu_2O_3 from Research Chemicals. These powders were weighed to the nearest tenth or hundredth of a gram and were then blended together in a polycarbonate plastic divider/mixer for three to four hours. This mixture was then charged via a quartz funnel into the crucible which was at a temperature to cause pre-reaction and sintering of the powdered components. Upon completion of the sintering of the full charge (usually accompanied by partial melting of the charge in the bottom of the crucible) the crucible temperature was raised to above the mixture's melting point where it was held for 24 to 48 hours to ensure complete mixing. Radial flow lines were clearly visible once the melting point of the charge was reached.

2.2 Czochralski Technique

The standard set-up for this series of growth runs is shown in Figure 3. A 3 inch by 3 inch iridium crucible, capped with a 2 inch I.D. iridium lid was used for most of the runs. The crucible is heated by RF coils powered by a 30 kw Level generator. The cover atmospheres for each run are detailed in the listing of experimental results (Table 1). The gases used (N_2 /forming gas or N_2 alone) were fed through holes centered in the steel table and zirconia base plate. The gas then exited from vertical slots cut in the zirconia support cylinder to flow up through the groq. The flow rates were generally in the range of 25 liters/min. The pulling rates and

Figure 3 RF Growth
Furnace Schematic



rotation rates are detailed in Table 1. Melt drop compensation was employed during runs where a boule was pulled.

2.3 Seed Generation

Bootstrapping seeds from previous growth runs has proven to be most successful in generating new single crystal growth from subsequent melts.

The first series of seeds were cut from spontaneously crystallized melts in 1.5 inch by 1.5 inch crucibles. These were used to grow small boules which yielded larger seeds. In the current run series all boules were grown from seeds mined from boules. The use of a 3 inch by 3 inch crucible and cooling the boule in contact with the melt has also produced sizeable sections of single crystal material from the frozen melt. Seeds have generally been suspended from an alumina rod on an iridium wire hook, although this has sometimes caused mechanical stability problems during growth. These larger single crystal pieces yield larger seeds, which will lead to improved growth. When suitably sized material is grown and large enough seeds can be manufactured, then a more rigid seed platform will be employed. This will reduce mechanical instabilities and facilitate better crystal growth. Some attempts at alternate orientations based on growth morphology and cleavage planes have been attempted, but the best growth direction appears to be parallel to the main cleavage plane. X-ray orientation of the seeds will also be used to more accurately determine the preferred growth orientation. Figure 4a shows a crystallized Gd_2SiO_5 melt while figure 4b shows a small seed crystal mined from a crystallized melt. Figure 4c shows some of the seeds mined from boules and used in later growth runs.



Figure 4a Crystallized Melt of Gd_2SiO_5 cooled from $1900^{\circ}C$ in a 2" x 2" iridium crucible



Figure 4b Original seed crystal of Gd_2SiO_5 culled from frozen melt

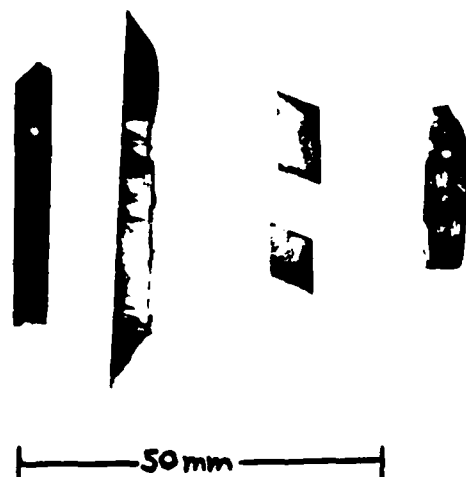


Figure 4c Sample of larger Gd_2SiO_5 seeds used for later boule growth

2.4 Polycrystalline Preparation

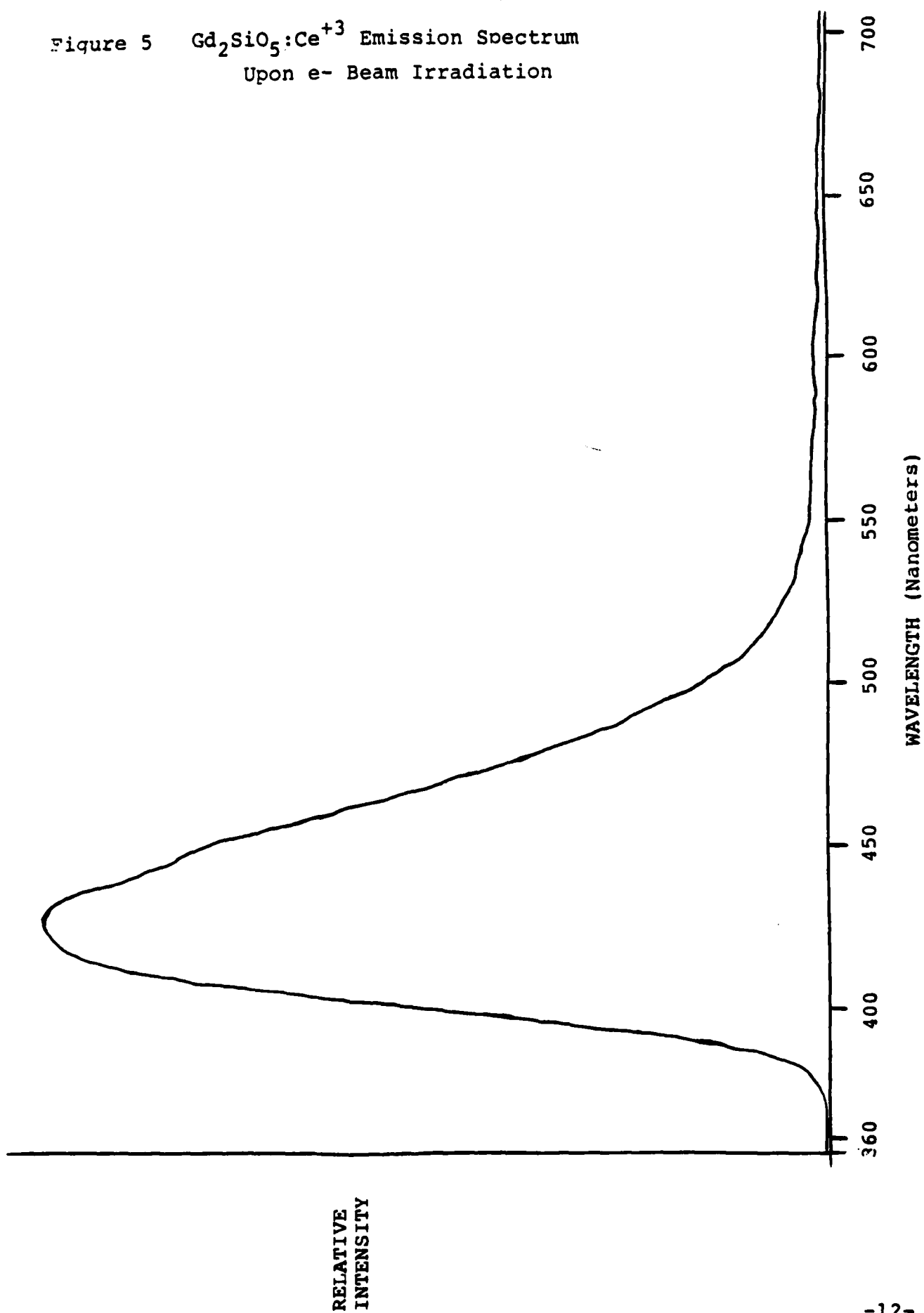
During this reporting period, it was determined that a quick survey of the potential for shifting the Ce^{+3} emission spectrum using compositional variations was in order. This was to be accomplished by a partial substitution of similar rare earths for the Gd in the Gd_2SiO_5 structure. Ten percent substitution was chosen as sufficient to cause an observable effect and this was carried out by using the precursor materials listed in the feed materials section of this report.

The samples weighed approximately 0.22g each and were pressed into aspirin sized tablets in a stainless steel die. The pellets were then sintered on the bottom of an iridium crucible to produce the orthosilicate structure. Then they were individually crushed and spectrally analyzed for their Ce^{+3} emission peak as outlined in the next section.

2.5 Cathodoluminescent Evaluation

Preliminary evaluations for luminescence were done under UV light. If a sample exhibited a visible response then a small single or polycrystalline sample was prepared for e- beam excitation. Upon exposure to the e- beam, the sample luminesces due to its Ce^{+3} content and this emitted radiation is scanned by a very accurate monochrometer and its intensity versus wavelength profile (from 300 nm to 700 nm) is recorded on a strip chart. This allows evaluation of the relative strength and peak wavelength position of the Ce^{+3} emission spectrum. A typical spectrum for $\text{Gd}_2\text{SiO}_5:\text{Ce}^{+3}$ is shown in Figure 5.

Figure 5 $\text{Gd}_2\text{SiO}_5:\text{Ce}^{+3}$ Emission Spectrum
Upon e- Beam Irradiation



2.6 Optical and Laser Evaluation

Optical and laser evaluation of samples from successful runs were to be performed at NRL. Samples for this purpose were to be prepared at Airtron by cutting single crystal sections from each boule and polishing them to optical clarity.

3.0 Results and Discussion

Since it appears that all the rare earths (R) form an isomorphous series in the compounds R_2SiO_5 , the crystal grower is presented with tremendous latitude in the preparation of both hosts and dopants for laser application. Let us consider the host crystal first. The elements La, Gd, Y and Lu span the full range of rare earth radii. Furthermore, each of these elements or their mixtures have trivalent ions with half or full shell electron complements; thus they are fully transparent in the absence of defect chemistry. The choice of Gd_2SiO_5 , which was grown first, is in the middle of the rare earth series. Doping with Ce^{3+} gives emission near 450nm. If Gd is mixed with Y, the emission should shift more towards the blue. If one mixes La and Gd, the peak emission may shift towards the red. By tailoring the type of mixed crystal, it should be possible to shift Ce^{3+} emission to the range of interest. Combinations of La-Gd, La-Y, La-Lu may be necessary.

The large size of the Ce^{3+} generally leads to a low distribution coefficient (k) in Y, or Lu analogs. The values are sometimes as low as 0.1-0.2. For Gd_2SiO_5 , k increases to about 0.4-0.5. If the La_2SiO_5 analog can be made, the k values are very close to 1. However, with the low dopant levels of Ce^{3+} (~0.1%) which are needed, there is not much difficulty in the phase behavior or crystal growth. We fully expected that Ce^{3+} could be introduced in low amounts (0.1%) in all crystals such as La, Gd, Y, or Lu compounds and so far this has been borne out.

Seed orientation plays a large part in both the crystal growth and the resulting laser properties. For noncubic materials, there may be serious reasons for investigating the optimum direction for growth. Similarly the polarization properties of emitted light are a topic for their efficient use in any laser. It has also been reported that the R_2SiO_5 type silicates have a possible plane of separation, and this has been observed in our growth runs. However, this does not seem to pose a problem since large single crystal sections have been grown.

Our preliminary effort concentrated on the growth of large single crystals, doping, orientation, crystal structure effects, and finally the mechanical processing of any grown crystals. The primary objective was the growth, but closely tied to this were physical studies on the absorption, fluorescent emission, lifetime, polarization, and other laser parameters. Most of the latter is to be done at the Naval Research Laboratory.

3.1 Phase Diagrams

A thorough examination of the available relevant phase diagrams (from the A.C.S.'s "Phase Diagrams for Ceramists" series) for the $R_2O_3-SiO_2$ systems (where R is a rare earth) was performed during this report period. All the phase diagrams exhibited similar structure especially around the one to one molar percent ratio line. The existence regions for the R_2SiO_5 compositions are bounded by solidus and liquidus curves and two eutectics around the 1:1 line. This can be seen in Figure 1 of the introductory section to this report. The temperature ranges of these existence regions are also very closely centered around $1900^\circ C$. This would indicate that R_2SiO_5 single crystal material should be isostructural as R is varied allowing mixed composition single crystal boules to be pulled stoichiometrically from the melt.

3.2 Doping and Atmosphere Control

The crystals grown in this series of experiments were doped by adding CeO_2 at about the 0.1% level of the precursor powder mixture. Conversion from Ce^{+4} to Ce^{+3} is thought to have been incomplete in the early growth runs. This was revealed by a yellowish colouration in portions of the melt and crystals. There was, however, always sufficient conversion to cause a very bright luminescence under UV illumination. More complete conversion appeared to be achieved in later runs by going from a flowing atmosphere of N_2 and forming gas to only a flowing N_2 gas cover. This was evidenced by the growth of transparent water white boules.

3.3 Gd_2SiO_5 Experimental Runs

The basic experimental conditions of each of the growth runs performed during this period are outlined in Table 1. Gd_2SiO_5 growth runs were attempted in experiments 10 through 16 and 18. Run 10 was an attempt to grow large enough single crystal sections to cut up for seeds. The result was a mostly single crystal boule with a strong yellowish cast which yielded four 30 mm long seeds. A seed from run 10 was used to initiate growth in run 11 which yielded a boule about 120 mm long but of widely varying diameter (from 8 to 28 mm). This boule had high surface stresses indicated by crazing of the surface and also exhibited a yellowish colour. Run 12 was seeded with material from run 10. Boule growth was eccentric and in trying to keep the boule from wobbling the growth axis was slowly shifted until there was an internal 130° angle between the longitudinal axis of the seed and the boule's longitudinal axis. The boule itself was mostly single crystal with some yellow colour being 22 mm in diameter by 57 mm long. Run 13 was an attempt to grow along the same axis as the boule tended to in the previous run using seeds cut from run 12's boule. This proved unsuccessful as the seeds kept burning off so the run was terminated. Run 14 was aborted after three days due to leakage in the crucible bottom. Another seed from run 10 was used to initiate growth in run 15. The seed was longitudinally bicrystalline and this propagated into the boule. Other than this, the boule was of water white transparency and highly single crystal. The lack of yellowish colour in the boule may be due to the absence of forming gas during the run. The same seed as was

used in run 15 (split to eliminate its bicrystallinity) initiated growth in run 16. The resulting boule showed an oblate cross-section (21/25mm) but was only grown to about 56mm in length due to arcing of the RF coils to the quartz cylinder. Run 18 was a test of the feasibility of growth of Gd_2SiO_5 along a different axis from those tried previously. Seeds were cut from the boule grown in run 16 at an orientation approximately 45° from both longitudinal and radial axes of the boule. This was parallel to regular fracture lines observed in the boule and thought to be crystallographic fracture planes. The boule grown at this orientation was mostly single crystal of water white transparency, but there was brittle fracture and extensive surface crazing which may have been due to the new growth direction. A photograph of sections of boules (oriented head up) are shown in Figure 6. From left to right they are from runs 11, 15, 16, and 18.

3.4 Polycrystalline Experimental Run

The luminescence results from run 17 are given in Table 1. Four samples were made up: one with all Gd_2SiO_5 as a reference, and three others, each with 10% substitution of Gd by Lu, Y, and La respectively. The average luminescence was about the same in all four samples as was the shape of the main Ce^{+3} emission peak. The secondary peak in the spectrum (observed at about 540 to 545nm) is suspected to be due to alternate site placement of the Ce^{+3} , but did not appear to shift in concert with the main peak. In fact, it almost disappeared into the slope of the primary peak when the primary peak shifted up in wavelength with the 10% Lu substitution. These results indicate that the pellets' crystal structures are basically identical and orthosilicate in form since the sintered $\text{Gd}_2\text{SiO}_5:\text{Ce}^{+3}$ spectrum matched the single crystal $\text{Gd}_2\text{SiO}_5:\text{Ce}^{+3}$



Figure 6 Boule Sections from (left to right)
Runs 11, 15, 16, and 18

luminescence pattern with peak emission at the same wavelength. These shifts were small however and went against the expected shift direction which would have the substitution of Y or Lu shift the luminescence peak to the blue and La substitution shift it to the red. No explanation of this reversal was known at the time of this writing although it appears to be a function of the polycrystalline preparation as is shown in Section 3.5.

3.5 Y_2SiO_5 and La_2SiO_5 Experimental Runs

Run 19 was an attempt to crystallize from the melt $Y_2SiO_5:Ce^{+3}$ using a Gd_2SiO_5 seed. This would allow verification of the crystal structure of $Y_2SiO_5:Ce^{+3}$ with X-ray analysis and comparison of its luminescence to that of $Gd_2SiO_5:Ce^{+3}$. The melt was allowed to mix by convection for about 48 hours, the Gd_2SiO_5 seed was dipped in, allowed to equilibrate and the melt was set for a programmed cool-down. The frozen melt's surface showed a crystallization pattern emanating from the seed contact site, but core drilling revealed incompletely mixed material in the lower portions of the crucible mass. This may explain why the X-ray results indicate the possible presence of more than one phase and don't correlate with the X-ray patterns observed for Gd_2SiO_5 to date. An island dipped from the melt prior to seeding was analyzed using the e- beam luminescence system and showed a peak emission at 414nm. Although this is more in line with the published data, the result is suspect due to the incomplete mixing of the melt. Run 20 was performed using $La_2SiO_5:Ce^{+3}$ in a manner similar to run 19, but even worse segregation was obvious upon core drilling. Both these experiments will be repeated in future runs with better mixing.

3.6 Cathodoluminescence Results

Samples from all the runs which yielded boules and runs 17, 19, and 20 exhibited bright luminescence from their Ce^{+3} content when exposed to UV radiation. Measurements using the e- beam cathodoluminescence apparatus gave luminescence spectrums showing the typical broad band Ce^{+3} profile. Shifts were observed which have been tentatively assigned to compositional variations, but these have to be confirmed with further single crystal growth runs.

3.7 Optical and Laser Evaluations

Samples from runs 10, 11, 12, 15 16, and 18 were sent to the NRL in or shortly after the period covered by this report. Results from NRL tests will indicate whether these materials are suitable candidates for laser substrates.

Table 1
List of Experimental Runs

RUN #	IRIDIUM CRUCIBLE SIZE	COVER GAS	PULL & ROTATION RATES	RUN DURATION	OTHER RUN PARAMETERS	RUN RESULTS
10	1.5" X 1.5	Forming & N ₂	.05"/Hr.	22 days	seed suspended on Ir wire	Seed kept burning off, boule growth terminated by burn off, yellow translucent single crystal boule 17mm diameter by 45mm long.
11	3" X 3"	"	"	8 Days	"	Irregularly shaped yellowish boule, highly stressed mostly single crystal, leading section of boule 14mm diameter by 55mm long, trailing, section flared from 8mm dia- meter to 28mm over a length of 67mm.
12	"	"	.05"/Hr. 45 RPM	22 Days	"	Boule growth was eccentric and in trying to keep the boule from wobbling the growth axis shifted by about 50° from the vertical, the boule was mostly single crystal with some yellow color and 22 mm diameter by 57mm long.

Table 1 (continued)

RUN#	IRIDIUM CRUCIBLE SIZE	COVER GAS	PULL & ROTATION RATES	RUN DURATION	OTHER RUN PARAMETERS	RUN RESULTS
13	3" X 3"	Forming & N ₂	45 RPM	27 Days	seed suspended on Ir wire seed at same orientation as boule grew at in run #12	Seed burn off prevented boule growth - run aborted when all of seed gone.
14	"	N ₂		3 Days	Power off twice to switch control boards	Crucible bottom leaked possibly due to power shut- downs.
15	"	"	48 RPM .0075"/Hr.	16 Days	Seed suspended on Ir wire	Boule come out bicrystalline due to seed, highly single crystal otherwise, water white transparency, some cracking, boule diameter went from 10mm at the seed to 30mm at a length of 34mm and maintained this diameter to the 55mm mark.
16	"	"	"	15 Days	Some seed as for run #15 but split in half to eliminate bicrystallinity	Oblate boule with cracking, mostly single crystal, water white transparent, straight portion of boule about 44mm long with an oblate cross section of 21-25mm.

Table 1 (continued)

RUN #	IRIDIUM CRUCIBLE SIZE	COVER GAS	PULL & ROTATION RATES	RUN DURATION	OTHER RUN PARAMETERS	RUN RESULTS
17	2" X 2"	N ₂		1 Day	Four pressed powder pellets of varying composition sintered on the bottom of crucible	<p>Pellets crushed and analyzed for e-beam fluorescence, excitation, the results:</p> <p>Gd₂SiO₅: Ce⁺³ = 429nm pk. (Gd_{0.9}La_{0.1})₂SiO₅: Ce⁺³ = 427nm " (Gd_{0.9}Y_{0.1})₂SiO₅: Ce⁺³ = 432nm " (Gd_{0.9}Lu_{0.1})₂SiO₅: Ce⁺³ = 435nm "</p>
18	3" X 3"	"	48 RPM .025"/Hr.	39 Days	Seed cut from and oriented 45° from radial and longitudinal axes of boule from run #16	<p>Crystal started to spiral crack on lifting from melt, therefore cooled in contact with melt, badly frosted surface due to surface crazing, bulk cracked also but water white transparent mostly single crystal, large pieces of single crystal in melt due to contact with boule during freeze out.</p>
19	"	"		7 Days	Y ₂ SiO ₅ :Ce ⁺³ melt Gd ₂ SiO ₅ Seed	<p>Melt allowed to cool in contact with seed, produced radial crystallization pattern, core drilling of crucible mass shows segregation in lower portions, X-ray diffraction results indicate multiple phases none of which correlate with Gd₂SiO₅ patterns.</p>

Table 1 (continued)

RUN#	IRIDIUM CRUCIBLE SIZE	COVER GAS	PULL & ROTATION RATES	RUN DURATION	OTHER RUN PARAMETERS	RUN RESULTS
20	3" X 3"	N ₂			La ₂ SiO ₅ :Ce ⁺³ melt Gd ₂ SiO ₅ seed	Melt allowed to cool in contact with seed, some large single crystal areas but core drilling revealed large segregation in lower reaches of crucible mass, left out overnight solid crucible mass turned to less dense powder (probably La ₂ O ₃), X-ray results of crystalline portion do not correlate with Gd ₂ SiO ₅ patterns.

4.0 Conclusions

The work performed so far under this contract has served to point out the most productive direction for further research. We now know that Ce^{+3} exhibits a very high luminescent activity in a wide range of R_2SiO_5 compounds when present at about the 0.1% level. We have also observed shifts in the Ce^{+3} cathodoluminescent spectrum due to compositional substitution of the rare earth component of the orthosilicate structure. The wavelength shift from $\text{R} = 100\% \text{ La}$ to $\text{R} = 100\% \text{ Lu}$ may be as much as 50 to 70nm. One objective now is to optimize compositions so as to produce the desired emission spectrum together with exploration of other orthosilicate systems which may broaden the available wavelength range.

Growth of small boules of Gd_2SiO_5 has been successfully undertaken during this report period. The goal now is to extend these growth techniques to produce larger, more perfect boules of Gd_2SiO_5 for laser rod fabrication. This can then be easily adapted to other orthosilicate systems for production of large boules from these materials as well.

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